

Electrical properties of a-antimony selenide

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This paper reports conduction mechanism in a-Sb₂Se₃ over a wide range of temperature (238K to 338K) and frequency (5Hz to 100kHz). The d.c. conductivity measured as a function of temperature shows semiconducting behaviour with activation energy $\Delta E = 0.42$ eV. Thermally induced changes in the electrical and dielectric properties of a-Sb₂Se₃ have been examined. The a.c. conductivity in the material has been explained using modified CBH model. The band conduction and single polaron hopping is dominant above room temperature. However, in the lower temperature range the bipolaron hopping dominates.

Keywords. Chalcogenides glasses; amorphous semiconductors; a.c. conductivity; electrical properties.

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I. INTRODUCTION

During the last few years amorphous semiconductors have been used in the manufacture of solar cells, batteries and phototransistors as well as in some steps of technological processing of very large scale integration (VLSI) microelectronics circuits [1]. Thermally induced structural and electrical effects are known to be important in inducing the memory phenomenon in semiconducting chalcogenide glasses [2]- [6]. Alzewel et al., [7,8] have studied the electrical conductivity of powdered antimony chalcogenides. Antimony alloys

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have received great attention in the past due to semiconductivity of Sb concentration. It was reported by Mott and Devis [9] that the effect of Sb in Se is even greater than in Te in promoting crystallisation. Recently, study has been made on thin and thick films of the system $Bi_{10}Sb_xSe_{90-x}$ ($x = 35, 40$ and 45) [10]. Crystalline Sb_2Se_3 has been prepared and studied by several workers [11,12]. Antimony triselenide (band gap ~ 1.2 eV), [13] a p-type semiconductor having an orthorhombic D_8^5 structure with unit cell dimensions $a = 11.77 \pm 1, b = 3.962 \pm 7, c = 11.62 \pm 14^0$ [14] is used in preparation of recording optical laser disc and as an insulating material in MIS solar cells. The preparation and study of amorphous- Sb_2Se_3 can throw light on the possible applications of this material in that form. Measurements of electrical properties of a- Sb_2Se_3 are reported in this paper over a wide temperature and frequency range. Results indicate that electrical properties can be explained on the basis of widely accepted modified-Correlated Barrier Hopping (m-CBH) model for chalcogenide glasses.

II. EXPERIMENTAL DETAILS

Ingots of glassy material Sb_2Se_3 were prepared by melt quench method. The 99.99% pure elements were sealed in an evacuated (1×10^{-6} torr) quartz ampoule (12mm diameter and 160mm long). The ampoule was heated in a muffle furnace at the reaction temperature (630 ± 3^0C for 24 hr). The composition (Sb:Se:1:1) of the glasses was determined by EDAX and amorphous nature was verified by X-ray diffraction pattern obtained using by X-ray diffractometer (Philips PW 1130/60). The pellets of 0.677 cm diameter were prepared in the hydraulic press at a pressure of $\approx 10^7 Pa$.

The measurements were carried out in a specially designed vacuum cell in the temperature range 233 to 313 K and frequencies from 6 Hz to 100 kHz. A sophisticated computer controlled ac-impedance system (Model 5206, EG & G, PARC USA) was used for the measurements [17]. Polished samples with a thickness 0.04-0.10 cm and area 0.36 cm^2 were put in contact with aquadag (a conducting emulsion) in a sandwich configuration. The sample

was kept in vacuum in a copper sample holder ensuring no temperature gradient between the electrodes and the sample. The effect of temperature is studied by using a closed cycle refrigerator (Model F-70, Julabo) was used to obtain different temperatures, which can maintain constant temperature within ± 1 K for all the measurements. Ohmic contacts were confirmed through linear I-V characteristics in the voltage range. All electrical measurements of real and imaginary components of impedance parameters (Z' and Z'') and real and imaginary component of admittance parameters (Y' and Y'') were made over a wide range of temperature (233 to 313 K) and frequency (6 Hz to 100 kHz).

III. RESULTS AND DISCUSSION

A. Dipolar behaviour of Sb_2Se_3

Figure 1 shows the plot of $\ln \sigma_{dc}$ versus $1000/T$ which was obtained using the expression

$$\sigma_{dc} = C' \exp(-\Delta E/kT). \quad (1)$$

The plots in Fig. 1 are found to be linear over the temperature range studied. In the above expression $C' = \sigma_0 \exp(\gamma/k)$ and γ is the temperature coefficient of the band gap. From the figure, the activation energy ΔE and C' for the sample are found to be 0.43 eV and $3.35 \times 10^{-4} \Omega^{-1} \text{cm}^{-1}$ respectively. It has also been observed that ΔE is constant for different pellets (thickness = 0.04cm to 0.08 cm) with a statistical error of 0.3%.

The variation of $\log |Z|$ versus $\log f$ (Fig. 2) indicates that the impedance is sensitive to frequency at lower temperatures and gradually becomes independent of frequency with increase in temperature. This type of behaviour indicates dominance of band conduction at higher temperatures.

The sensitivity of Sb_2Se_3 to temperature is shown in Fig. 3. The figure gives the Nyquist plots (Z' versus Z'') for Sb_2Se_3 at different temperatures over the frequency range studied. The semicircular Nyquist plots indicate dipolar nature of the sample, which may be due to hopping back and forth of bipolarons between charged defects states (D^+ and D^-).

reported to be present in chalcogenides [14]. The size of the Nyquist plots reduces with rise in temperature and regain their original size when temperature is increased.

The Nyquist plots for Sb_2Se_3 form a perfect arc of a semicircle with its centre lying considerably below the abscissa (i.e., distribution parameter $\alpha \gg 0$). Figure 3 also gives the value of α calculated at different temperatures. In dielectric materials, the finite value of the distribution parameter α and a depressed arc are typical for a dipolar system involving multirelaxation processes [14]- [17]. But the dependence of α on temperature indicates that the mean position of the dipole is varying with variation in temperature. The dipolar nature of Sb_2Se_3 is further confirmed by investigating the variation of capacitance with temperature at different frequencies. It is clear from Fig. 4 that in the lower temperature range the value of C is nearly constant and it increases with increase in temperature. However the rate of change of capacitance (i.e., dC/dT) is higher for lower frequencies and decreases with increasing temperature. These results can be explained by the fact that thermally assisted hopping results in increasing the capacitance of the material. In other words, in the lower temperature range, the dipoles remain frozen and attain rotational freedom when the temperature is increased. Thus the relaxation effects are confirmed by the rate of increase in capacitance with frequency.

The inversion of Nyquist plots in the admittance plane are shown in the Fig. 5. It is clear from the figure that the angle of inclination is less than $\pi/2$. This type of behaviour has been explained on the basis of R-C network model (Fig. 6), such that total impedance is given by [18]

$$\begin{aligned} Y(\omega) &= B(i\omega)^n + i\omega C_\alpha + G_0 \\ &= G_0 + B\{\cos(s\pi/2) + i \sin(s\pi/2)\}\omega^n + i\omega C_\alpha \end{aligned} \quad (2)$$

Moreover, this angle (shown in Fig. 5) is also temperature dependent. At lower temperatures (below 273 K) it is almost constant, while after a certain temperature (276 - 278 K) it shows a sudden decrease, which implies a change in the conduction mechanism.

B. CBH model for a-Sb₂Se₃

The complex impedance plots show the dipolar multi-relaxation nature of a-Sb₂Se₃. This behaviour is explained by the R-C network model (Fig. 6). However, this does not further elucidate the nature of conduction mechanism in this sample. Therefore a.c. conductivity $\sigma_{ac}(\omega)$ at different temperatures is calculated. It is found that dispersion in the temperature dependence of $\sigma_{ac}(\omega)$ curves increases at lower frequencies while it merges at higher frequencies (Fig. 7).

From the expression

$$\sigma_{ac}(\omega) = \sigma_{total}(\omega) - \sigma_{dc} \quad (3)$$

the a.c. conductivity is obtained, where σ_{dc} is given by Eqn.1 and its behaviour is shown in the Fig. 1. The $\sigma_{total}(\omega)$ is measured directly from lock-in-analyser (Section II). The $\sigma_{ac}(\omega)$ dominates over σ_{dc} at lower temperatures, while σ_{dc} dominates at higher temperatures (Fig. 7).

The modified Correlated Barrier Hopping (m-CBH) model explains the experimental results reported in this paper. It states that bipolaron hopping between the charged defect states D⁺ and D⁻ is responsible for the a.c. conductivity in these semiconductors. Dipolar multi-relaxation behaviour of this sample is already indicated from the Nyquist plots described in the last subsection. So the charged defect states are expected to be present in a-Sb₂Se₃. According to CBH model a.c. conductivity is given by [19,20]

$$\sigma_{ac} = \frac{n\pi^2 NN_p \epsilon' \omega R_\omega^6}{24} \quad (4)$$

where n is the number of polarons involved in the hopping process, NN_p is proportional to the square of the concentration of the states and ϵ' is the dielectric constant. R_ω is the hopping distance for the conduction $\omega\tau = 1$ and is given as

$$R_\omega = \frac{4ne^2}{\epsilon' [W_m - kT \ln(1/\omega\tau_0)]} \quad (5)$$

where W_m is the maximum barrier height and τ_0 is the characteristic relaxation time for the material.

Figure 7 shows the variation of a.c. conductivity with temperature at different frequencies. It is evident from the figure that σ_{ac} is very sensitive to temperature in the higher temperature regime. The low temperature a.c. conduction can be explained by considering bipolaron hopping between D^+ and D^- centers whereas the higher temperature behaviour is due to thermally activated single polaron hopping. At higher temperatures a number of thermally generated D^0 centers are produced with a temperature dependent concentration [21]

$$N_0 = N_T \exp(-U_{eff}/2kT) \quad (6)$$

where N_T is the concentration of D^+ or D^- centers at $T = 0K$. The defect concentration factor NN_P in Eqn. 4 is replaced by :

$$NN_P = N_T^2/2 \quad (\text{For bipolaron hopping}) \quad (7)$$

$$NN_P = N_T^2/2 \exp(-U_{eff}/2kT) \quad (\text{For single polaron hopping}) \quad (8)$$

Thus for single bipolaron hopping this factor is thermally activated and hence $\sigma_{ac}(\omega)$ is also activated.

According to CBH model [22]

$$B = W_m - W_1 + W_2 \quad (9)$$

where B is optical band gap, W_m is the maximum band width and behaviour with frequency ω is as

$$\sigma_{ac}(\omega) = A\omega^s \quad (10)$$

Here [[19]],

$$s = \frac{d(\ln \sigma_{ac})}{d \ln \omega} = 1 - 6kT/W \quad (11)$$

While $W = W_m$ for bipolaron hopping and $W = W_1$ or W_2 for single polaron hopping. Therefore, W_m is estimated from the s values at lower temperatures. Values of W_1 , W_2 and U_{eff} were estimated from the values of s at higher temperatures. These parameters are then fed to CBH model to fit the a.c. conductivity data calculated from the experiments. The behaviour of Eqn.11 ($s = 1 - 6kT/W^*$) can be seen in the Fig. 8. This indicates that W^* follows W_M for lower temperatures and W_2 for higher temperatures. Figure 8 also shows a sudden change in conduction mechanism at a certain temperature (276-278K). The similar behaviour is noticed in the admittance plots (Fig. 5). The different parameters used for CBH model calculations are as following

Further, the effect of temperature and frequency on the hopping length R_ω , which is a measure of the effective length of a dipole can be studied using CBH model. The variation of R_ω with frequency at different temperature is shown in the Fig. 9. The figure indicates that R_ω is more sensitive to frequency in the high temperature regime and the sensitivity decreases with decreasing temperature. Thus at low temperature, R_ω is constant with frequency, which implies a constant value of capacitance which is found to be true from the Fig. 4.

IV. CONCLUSION

Nyquist plots or complex impedance studies confirm the dipolar nature of Sb_2Se_3 and multi-relaxation behaviour as seen in most chalcogenide glasses. These plots also indicate that the capacitive nature dominates over the resistive nature of the sample at lower temperature (< 273 K), while the resistive nature is dominant at higher temperatures. The conduction mechanism has a sudden change from bipolaron to single polaron hopping (276-278 K). The temperature and frequency dependence of a.c. conductivity is well explained by the m-CBH model. The contribution to a.c. conductivity from single polaron hopping is dominant at higher temperatures.

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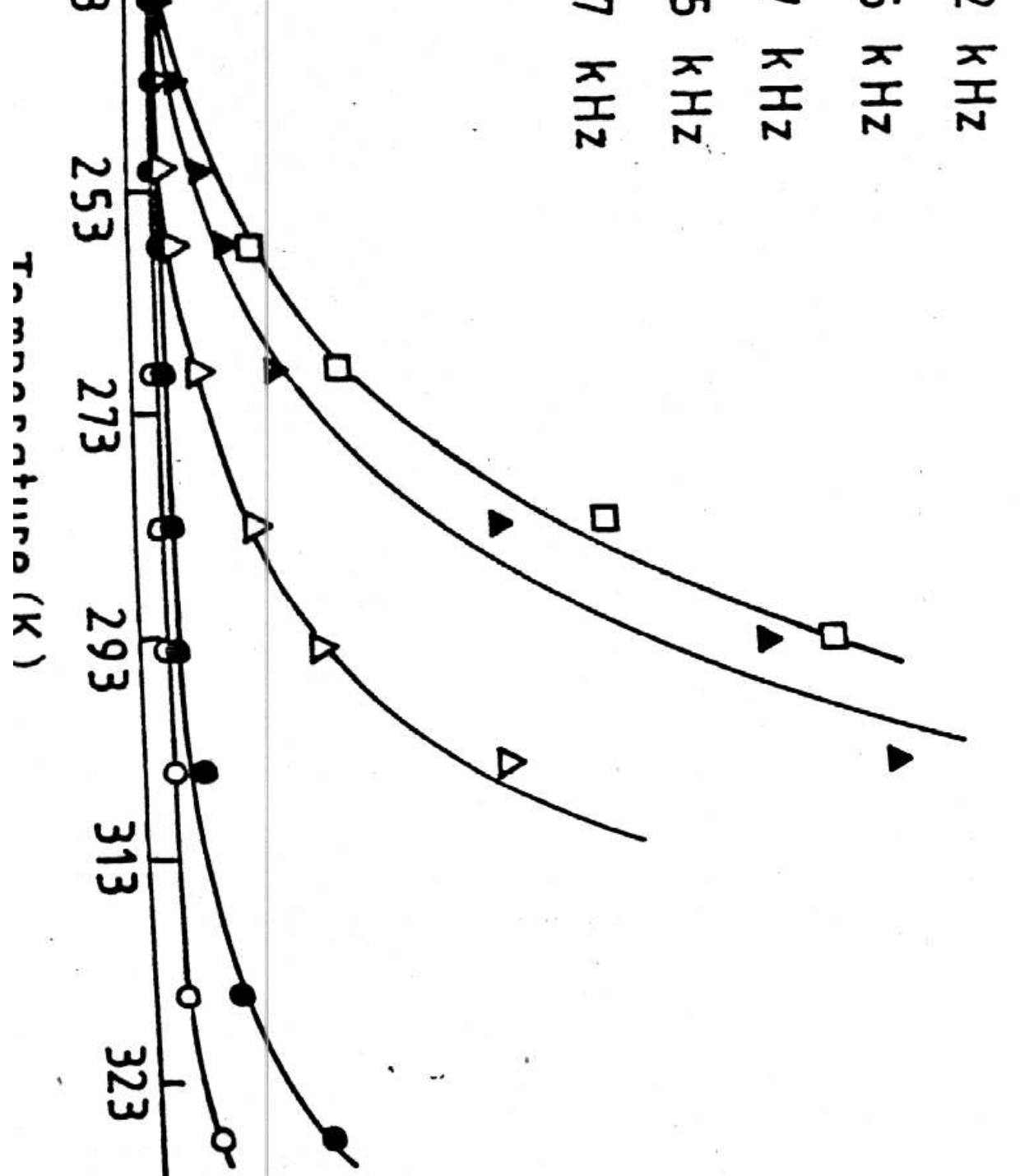


FIG. 1. Variation of $\sigma_{dc}(\Omega^{-1}cm^{-1})$ with respect to $1000/T(K)$.

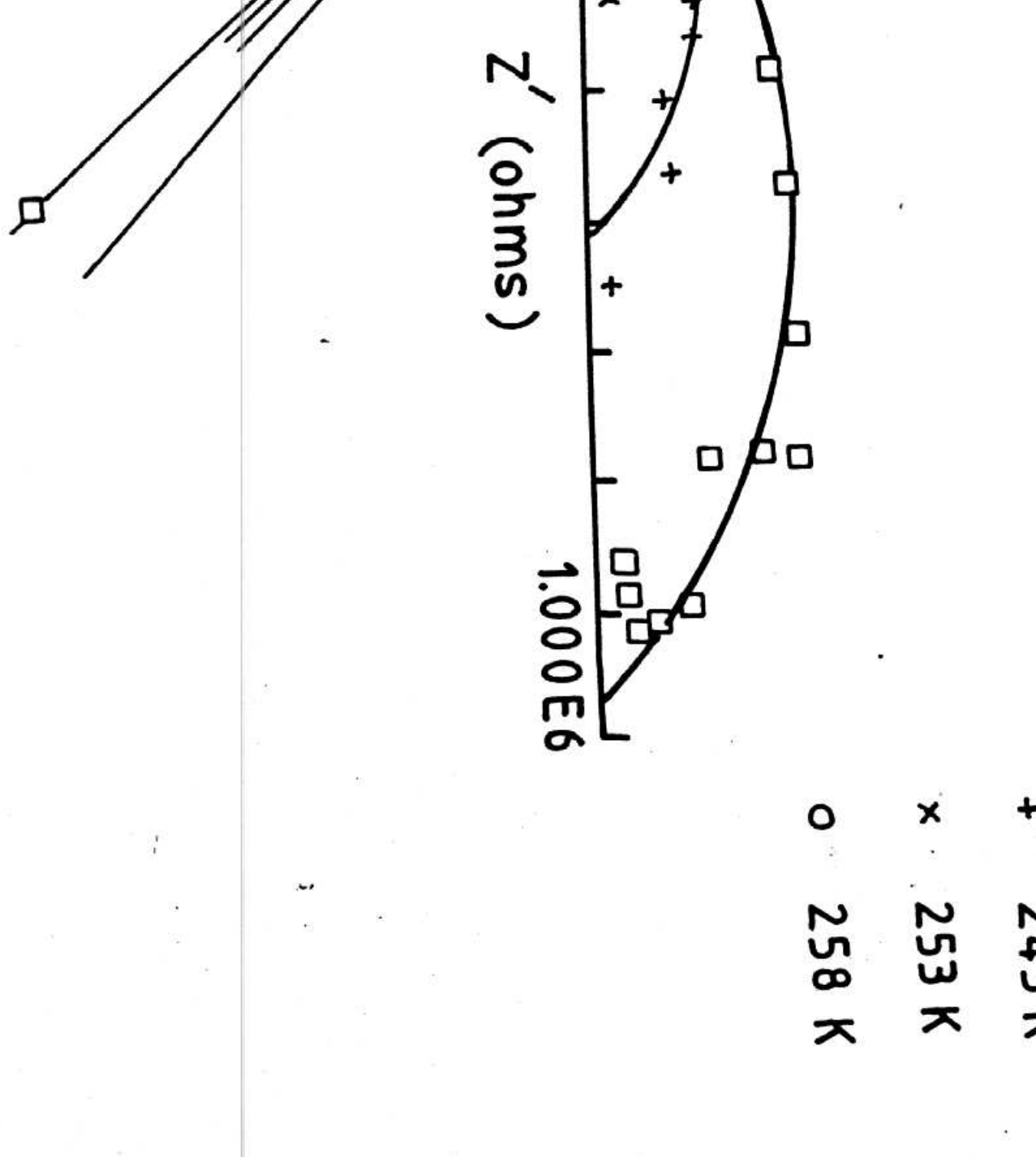


FIG. 2. Frequency dependence of $\log |Z|(\Omega)$ over the entire temperature range studied.

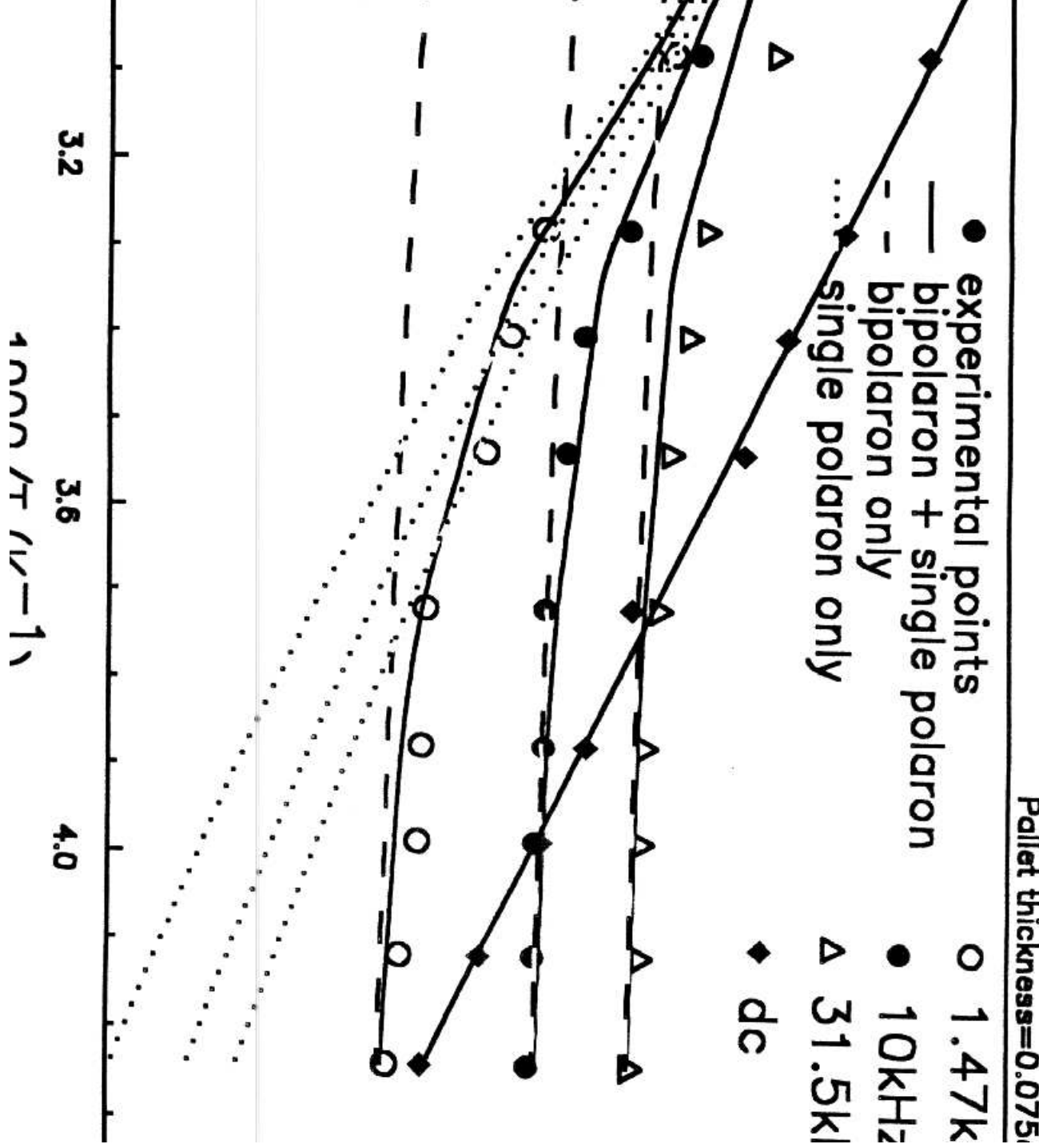


FIG. 3. Nyquist plots (Z' versus Z'') at different temperatures. Variation of angle $\alpha = (1-s)\pi/2$ with respect to temperature.

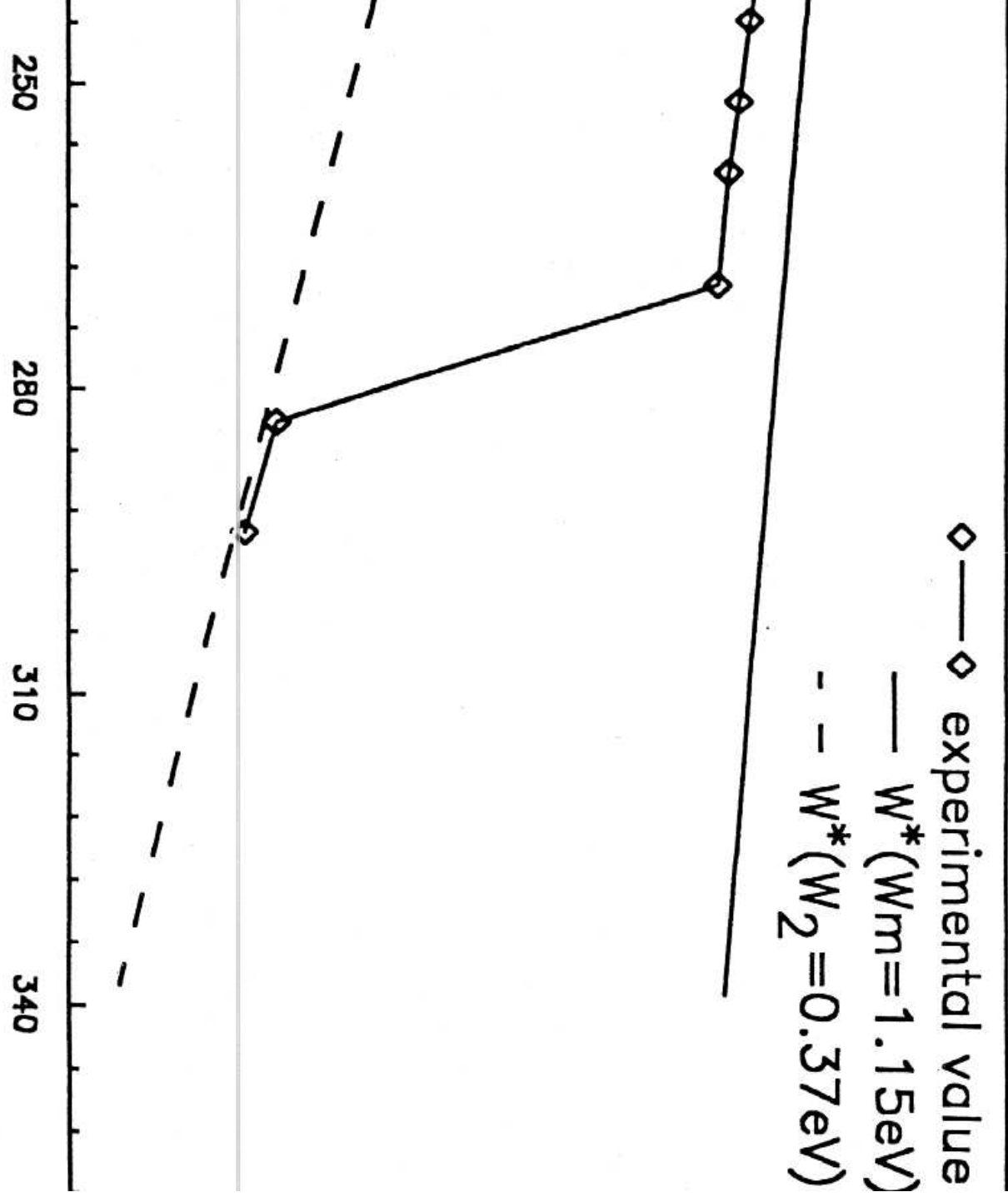


FIG. 4. Temperature variation of capacitance at different frequencies.

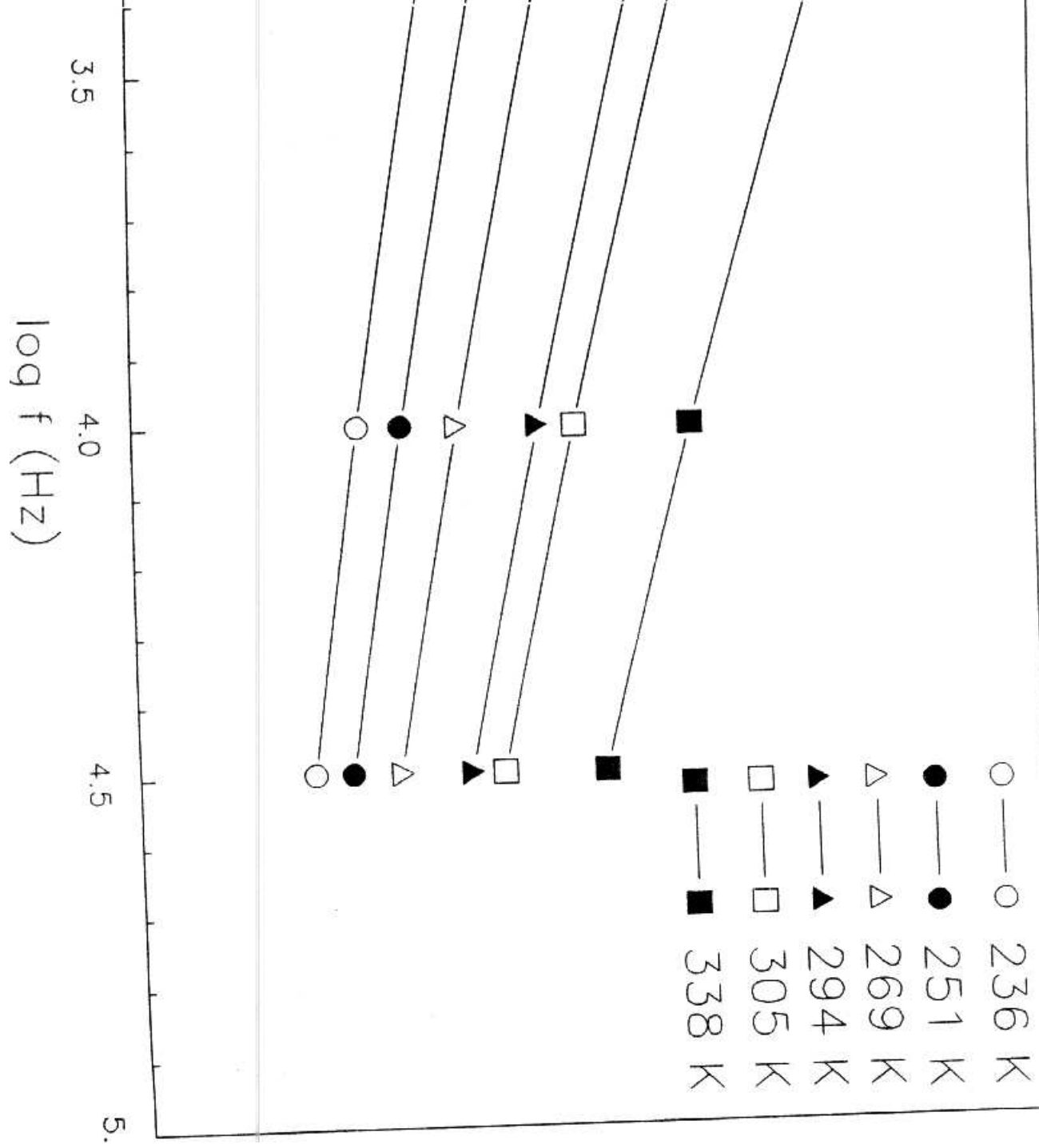


FIG. 5. Admittance plots ($Y'(\Omega^{-1})$ versus $Y''(\Omega^{-1})$). The variation of angle $\alpha = s\pi/2$ with temperature.

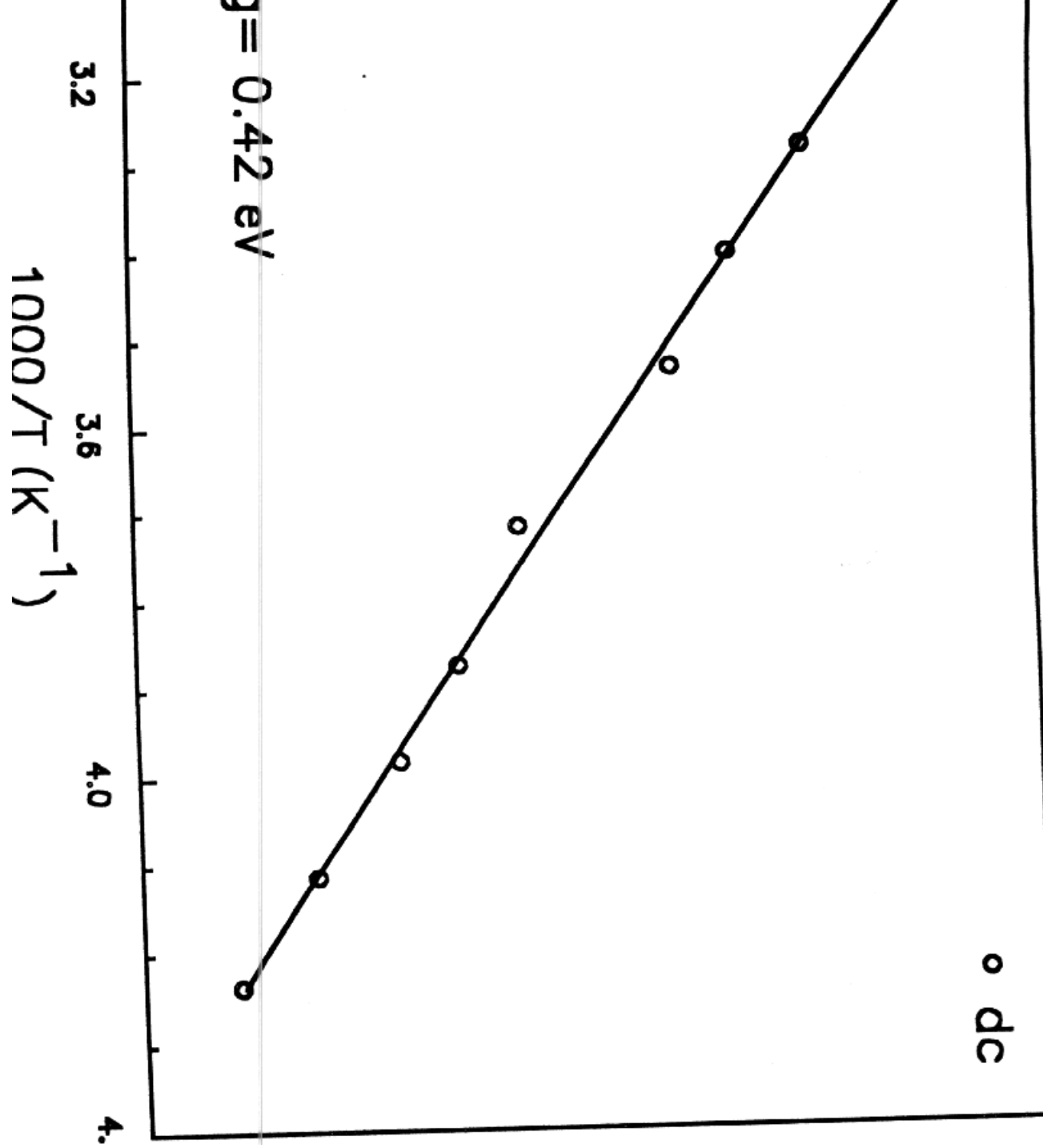


FIG. 6. R-C Network model for lossy capacitance.

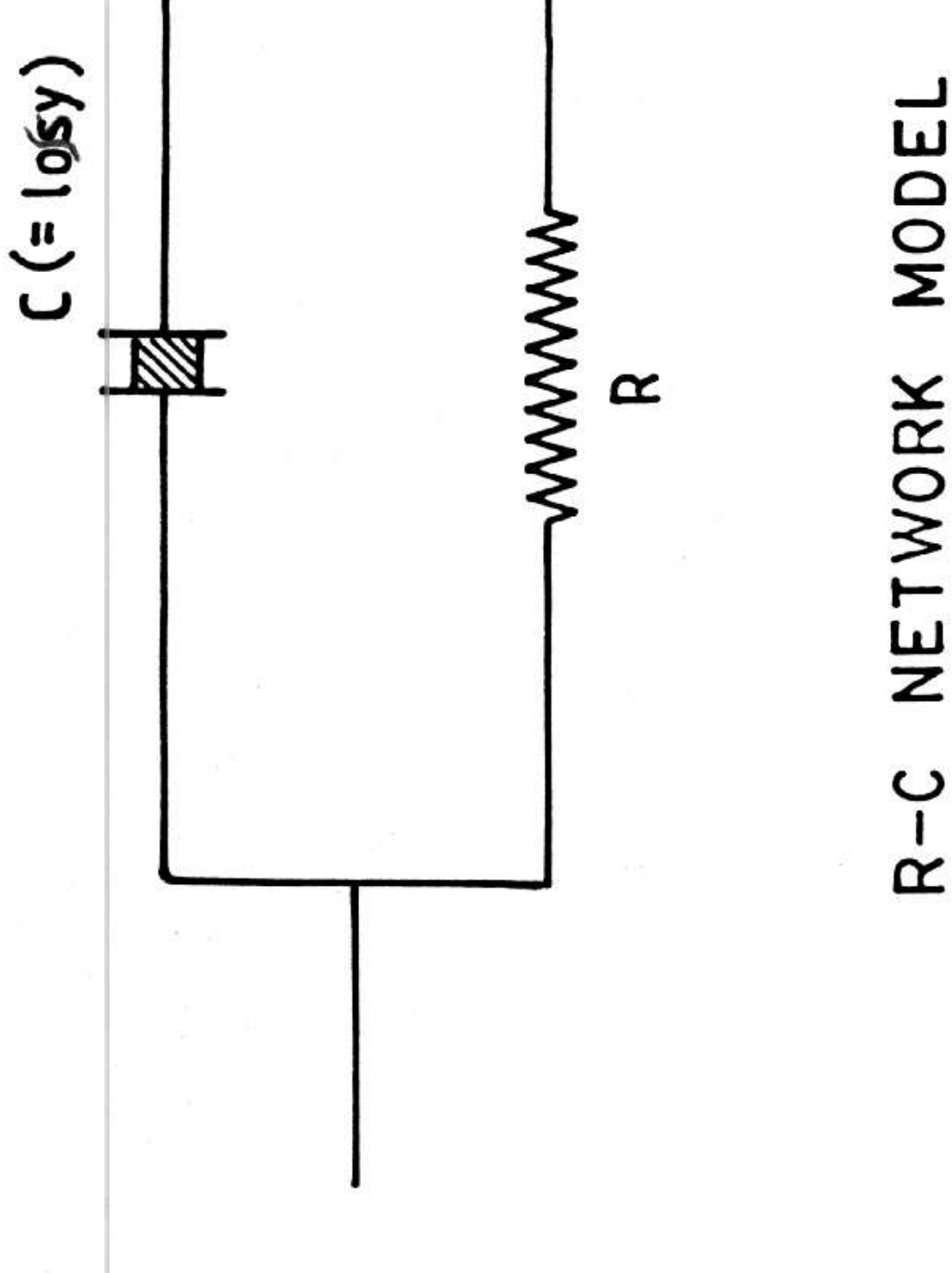


FIG. 7. CBH Fitting - The variation of $\sigma_{ac}(\Omega^{-1}cm^{-1})$ with respect to $1000/T$ for experimental points and theoretical curves (m-CBH) at different frequencies.

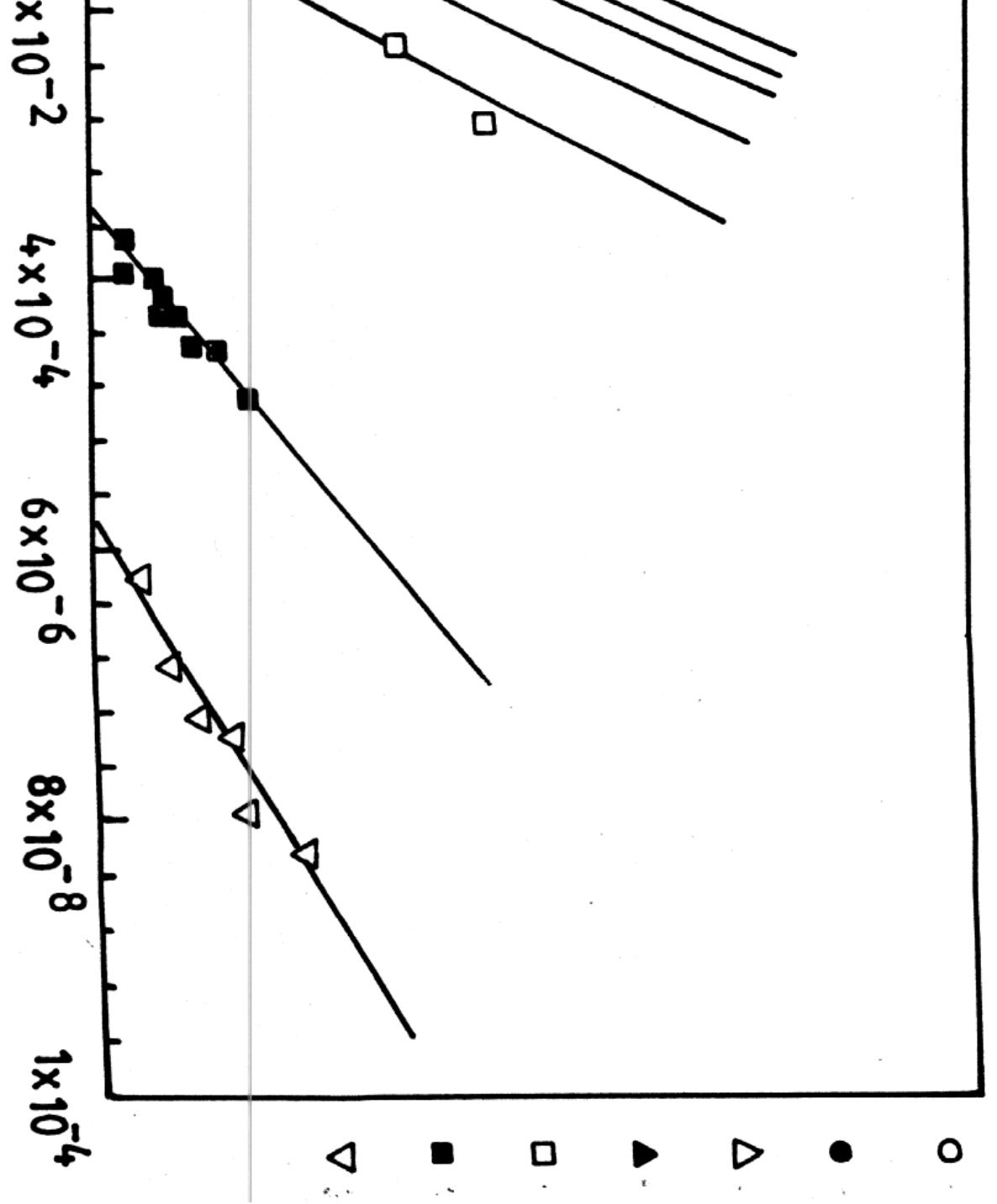


FIG. 8. Temperature dependence of $s(= 1 - 6kT/W^*)$ for theoretical curves (m-CBH) and experimental points.

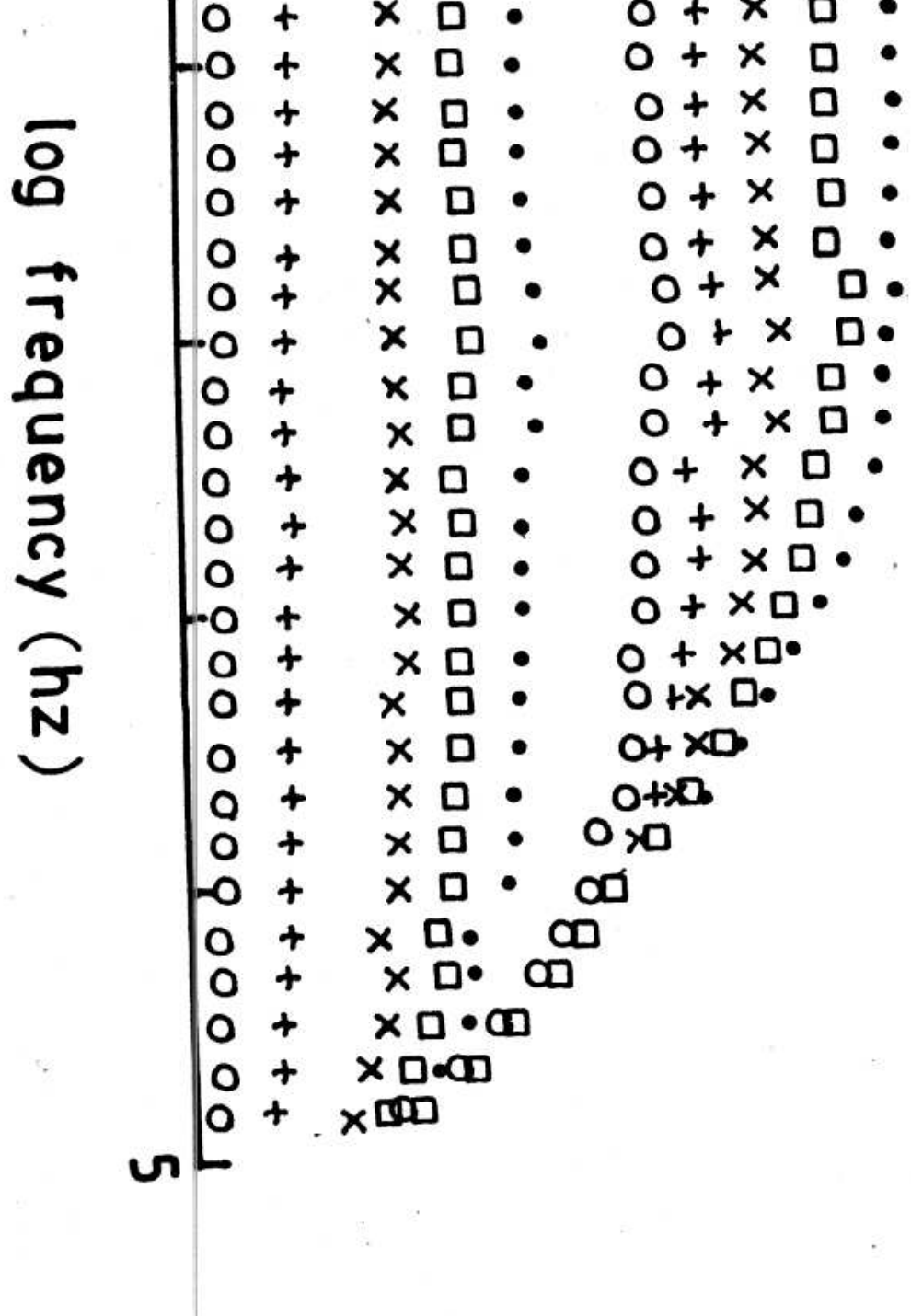


FIG. 9. Variation of $R_\omega(A^0)$ with respect to temperature (K) at different frequencies